

# Study on barium titanate ceramics prepared by various methods

M. C. CHEUNG, H. L. W. CHAN, C. L. CHOY

*Department of Applied Physics and Materials Research Centre, The Hong Kong Polytechnic University, Hungghom, Kowloon, Hong Kong, China*

*E-mail: apahlcha@polyu.edu.hk*

Barium titanate ( $\text{BaTiO}_3$ ) ceramics have been fabricated using powders prepared by sol-gel, coprecipitation and mixed oxide methods. The powders prepared by sol-gel and coprecipitation have average crystallite diameters of 100 nm and 300 nm, respectively while the diameter of the mixed oxide powder is  $\sim 1\text{--}3\ \mu\text{m}$ . When sintered at the same temperature of  $1320^\circ\text{C}$ , the three  $\text{BaTiO}_3$  ceramics have very different grain size, with the one prepared by the mixed oxide method having the largest grain size of  $20\ \mu\text{m}$ . The dielectric permittivity increases as the grain size of the ceramic becomes smaller. The room temperature ( $25^\circ\text{C}$ ) dielectric permittivity, pyroelectric and piezoelectric properties of these ceramics have been measured as functions of the poling field. The  $\text{BaTiO}_3$  ceramic fabricated from nanosized powder derived from the coprecipitation method is found to have the smallest grain size and better properties than prepared from the sol-gel route, and is thus a good candidate for use in devices that required thick (10 to  $20\ \mu\text{m}$ ) ferroelectric films. © 2001 Kluwer Academic Publishers

## 1. Introduction

Since the discovery of barium titanate ( $\text{BaTiO}_3$ ) in the early 1940's, there has been a continuous development of using it in new industrial and commercial applications.  $\text{BaTiO}_3$  is widely used as capacitors, ultrasonic transducers, pyroelectric infrared sensors and positive temperature coefficient (PTC) resistors [1–4]. Being a lead-free ferroelectric ceramic,  $\text{BaTiO}_3$  is an environmentally friendly material, thus making it a good candidate for various applications.

In order to prepare  $\text{BaTiO}_3$  ceramic thick films ( $10\text{--}20\ \mu\text{m}$ ) for use in various devices, the ceramic is required to have small grain size and reasonably good properties.  $\text{BaTiO}_3$  ceramic prepared from the conventional mixed oxide route has too large ( $> 10\ \mu\text{m}$ ) a grain size for thick film applications. In the present study,  $\text{BaTiO}_3$  powders prepared by the sol-gel process [5] and the coprecipitation method [6] are sintered at various temperatures to give ceramics with small grain size, the properties of which are then compared with those of  $\text{BaTiO}_3$  ceramics prepared by the mixed oxide route [7]. In the following sections,  $\text{BaTiO}_3$  (sol-gel),  $\text{BaTiO}_3$  (coprecipitation) and  $\text{BaTiO}_3$  (oxide) denote the ceramic samples prepared by sintering powder obtained from the sol-gel process, coprecipitation method and mixed oxide method, respectively. The dielectric permittivity, pyroelectric and piezoelectric properties of  $\text{BaTiO}_3$  ceramics prepared by various methods are investigated.

## 2. Sample preparation and experiments

Nanosized  $\text{BaTiO}_3$  powder was prepared either by the sol-gel process or by the coprecipitation method. In the

sol-gel process (Fig. 1), barium acetate and titanium isopropoxide were dissolved in deionized water and then mixed under continuous stirring at room temperature ( $25^\circ\text{C}$ ). The amounts of barium acetate and titanium isopropoxide were in a molar ratio of 1 : 1. In order to avoid precipitation, small amounts of acetic acid and acetylacetone were added. The sol was relatively stable and became a gel in a few days. The  $\text{BaTiO}_3$  gel was dried at  $100^\circ\text{C}$  for 24 h and then annealed at  $800^\circ\text{C}$  for 24 h to give  $\text{BaTiO}_3$  powder. The average crystallite size of the annealed powder was about 100 nm as determined by X-ray diffraction.

The  $\text{BaTiO}_3$  powder prepared by coprecipitation method was supplied by the Shanghai Institute of Ceramics. In the coprecipitation process, a mixed solution of barium chloride and titanium tetrachloride was added drop by drop into oxalic acid while being vigorously stirred. The amounts of barium chloride and titanium tetrachloride were in a molar ratio of 1 : 1. The amount of oxalic acid used was 20% in excess of the stoichiometric amount required to give precipitates. The resulting  $\text{BaTiO}_3$  precipitates were washed repeatedly in deionized water and then dried at  $80^\circ\text{C}$ . The crystallite size of the powder was about 300 nm.

The  $\text{BaTiO}_3$  powder prepared by the mixed oxide route was provided by Zhongshan University in Guangzhou. The  $\text{BaTiO}_3$  powder was prepared by reacting  $\text{BaCO}_3$  and  $\text{TiO}_2$  in air.  $\text{BaTiO}_3$  was initially formed at the  $\text{BaCO}_3\text{--TiO}_2$  grain boundaries. Then  $\text{BaTiO}_3$  reacted with  $\text{BaCO}_3$  to form  $\text{Ba}_2\text{TiO}_4$  until all the  $\text{BaCO}_3$  was exhausted. Finally  $\text{Ba}_2\text{TiO}_4$  reacted with the remaining  $\text{TiO}_2$  to form  $\text{BaTiO}_3$ . The size of the powder was between 1 to  $3\ \mu\text{m}$ .

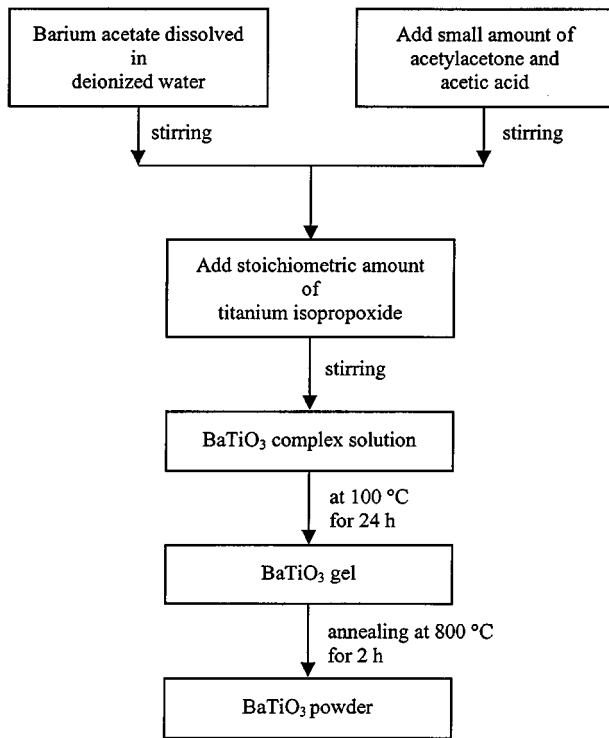


Figure 1 Flowchart showing the procedures for preparing BaTiO<sub>3</sub> powder by a sol-gel process.

The BaTiO<sub>3</sub> powders prepared by the above three methods were pressed into discs and sintered at various temperatures for 1 h to give the ceramic samples.

The structures of the samples were studied by scanning electron microscopy (Leica Stereoscan 440) and X-ray diffraction (Philips X'pert XRD System). For

electrical measurements, air-dried silver paint was applied to both sides of each sample to serve as electrodes. The relative permittivity  $\epsilon'$  and dielectric loss  $\tan \delta$  at 1 kHz were measured using a HP4194A impedance analyzer. After the BaTiO<sub>3</sub> samples had been poled at various d.c. fields, the pyroelectric coefficient  $p$  was measured using the digital integration method [8] and the piezoelectric coefficient  $d_{33}$  was measured by a Pennebaker model 8000 piezo  $d_{33}$  tester (American Piezo-Ceramic Inc.). A HP4194A impedance analyzer was used to measure the thickness electromechanical coupling coefficient  $k_t$  by a resonance technique following the IEEE standard [9].

### 3. Experimental results

#### 3.1. Structures and microstructures

Figs 2–4 show the SEM micrographs of BaTiO<sub>3</sub> (sol-gel), BaTiO<sub>3</sub> (coprecipitation) and BaTiO<sub>3</sub> (oxide), respectively. For the SEM measurements, the

TABLE I Showing the variations of grain size, tetragonality ( $c/a$ ) and density with sintering temperature for BaTiO<sub>3</sub> ceramics prepared by different methods

Synthesis method	Sintering temperature (°C)	Average grain size ( $\mu\text{m}$ )	$c/a$ ratio	Density ( $\text{kg/m}^3$ )
Sol-gel	1150	2	1.009	5574
	1200	5	1.010	5649
	1320	10	1.010	5685
Coprecipitation	1250	0.60	1.006	5250
	1280	0.72	1.007	5400
	1300	0.90	1.008	5510
	1320	1.13	1.009	5706
Mixed oxide	1320	20	1.010	5702

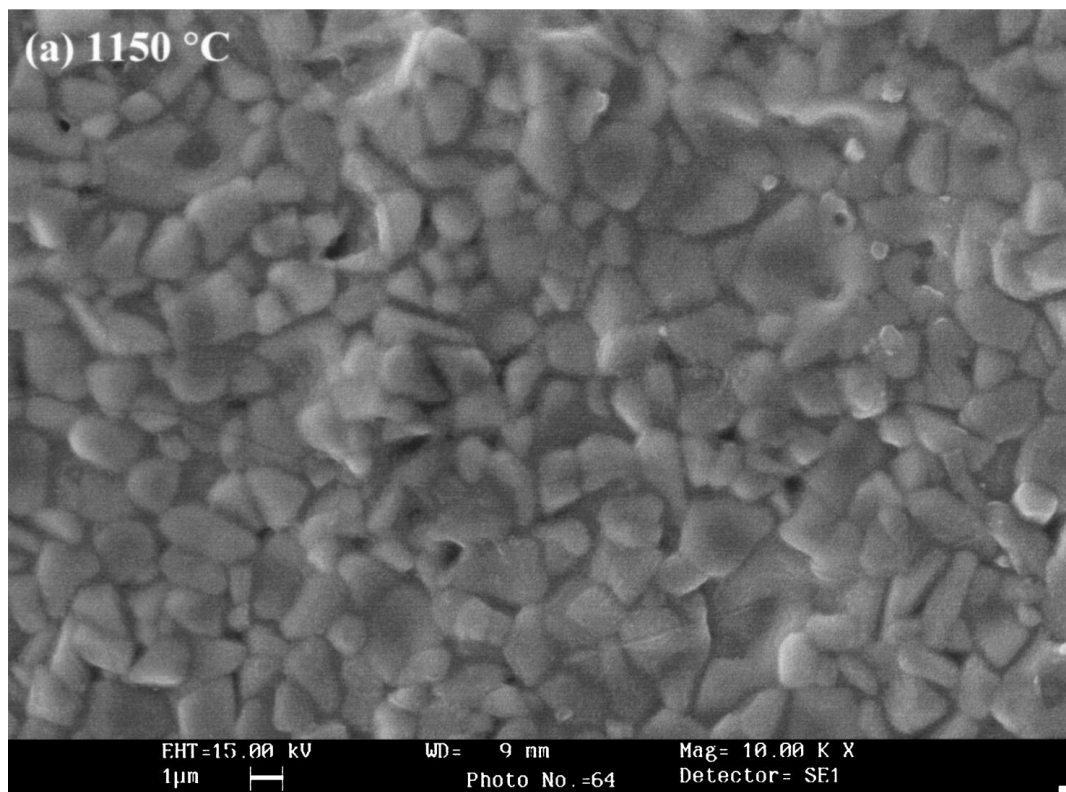


Figure 2 SEM micrographs of BaTiO<sub>3</sub> (sol-gel) ceramic sintered at (a) 1150°C, (b) 1200°C and (c) 1320°C. (Continued.)

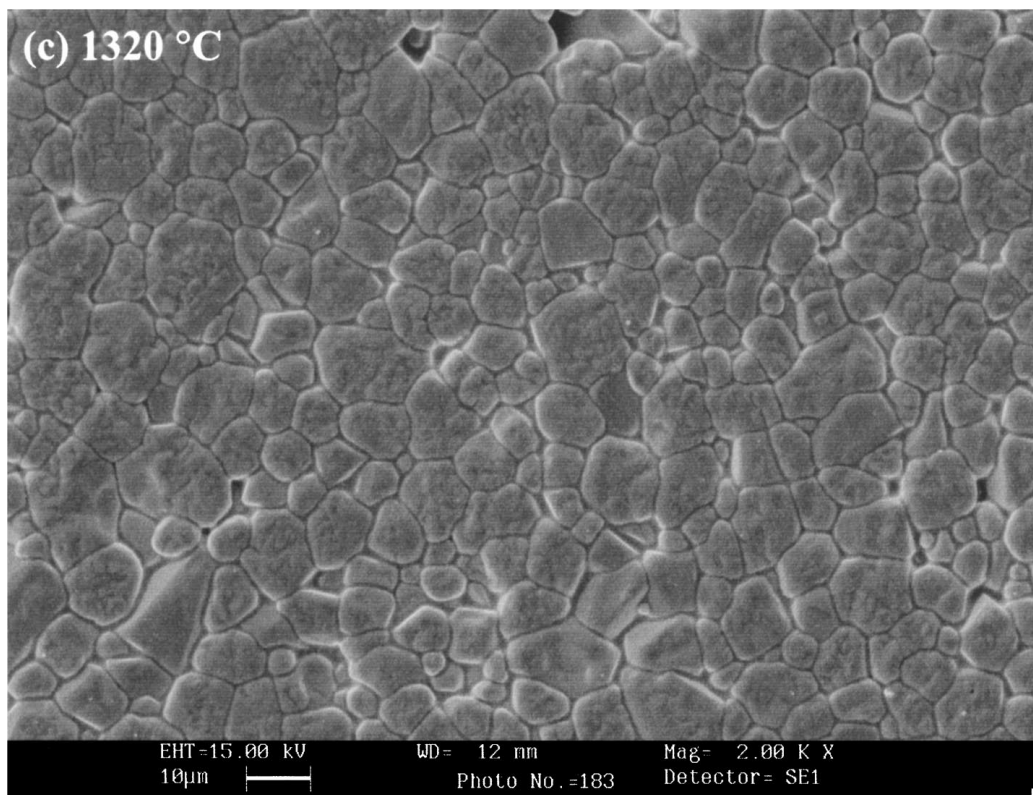
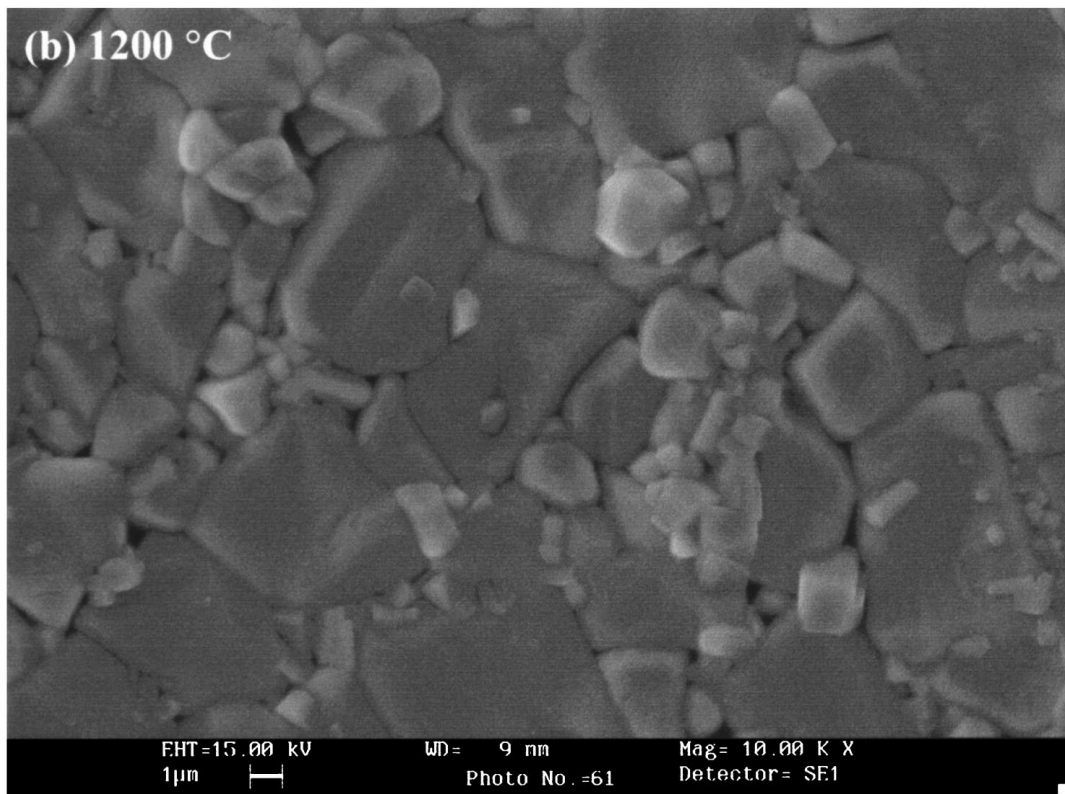


Figure 2 (Continued).

ceramics were polished and thermally etched at a temperature  $50\text{ }^{\circ}\text{C}$  below the sintering temperature. The average grain size was determined from the SEM micrographs and shown in Table I. For  $\text{BaTiO}_3$  (sol-gel), the grain size of the ceramic increases rapidly as the sintering temperature  $T_s$  increases, reaching  $10\text{ }\mu\text{m}$  at  $T_s = 1320\text{ }^{\circ}\text{C}$  (Fig. 2). However, the grain size of  $\text{BaTiO}_3$  (coprecipitation) is very small and shows

a much slower increase with sintering temperature, reaching only  $1.13\text{ }\mu\text{m}$  at  $T_s = 1320\text{ }^{\circ}\text{C}$  (Fig. 3). Following the conventional procedure,  $\text{BaTiO}_3$  (oxide) was sintered only at  $1320\text{ }^{\circ}\text{C}$ . This results in a grain size much larger than those in  $\text{BaTiO}_3$  (sol-gel) and  $\text{BaTiO}_3$  (coprecipitation) (Fig. 4).

Fig. 5 shows the XRD patterns of  $\text{BaTiO}_3$  (coprecipitation) sintered at  $1320\text{ }^{\circ}\text{C}$ . Although not shown,

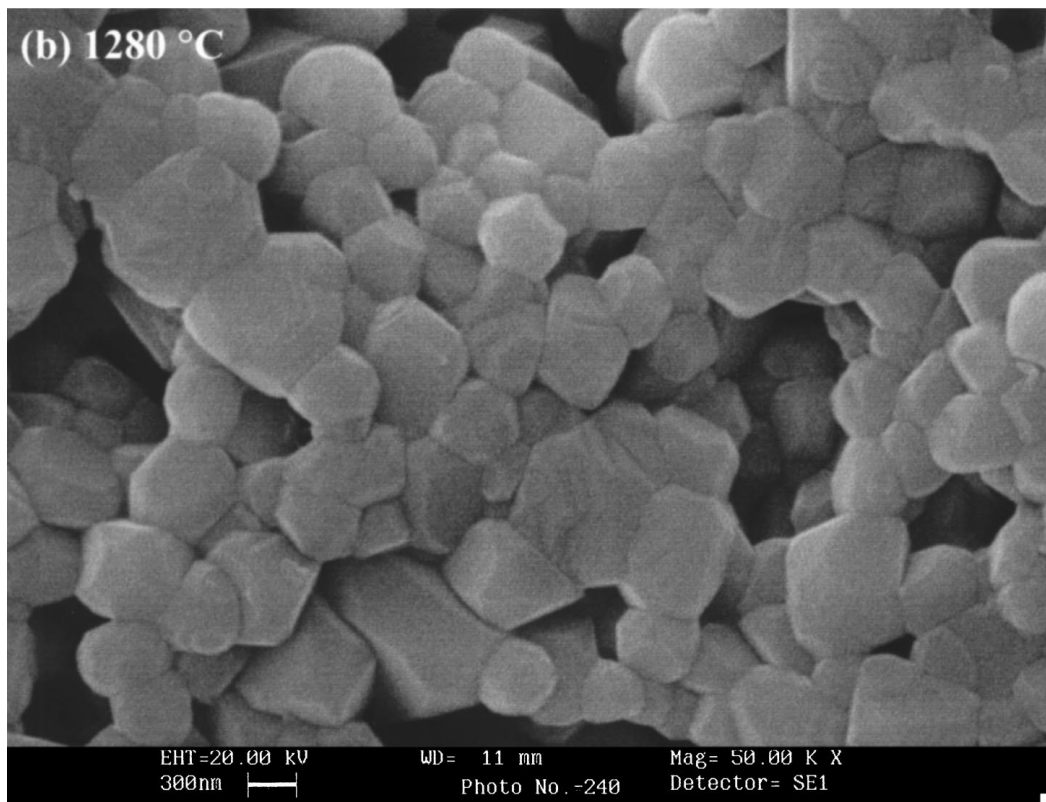
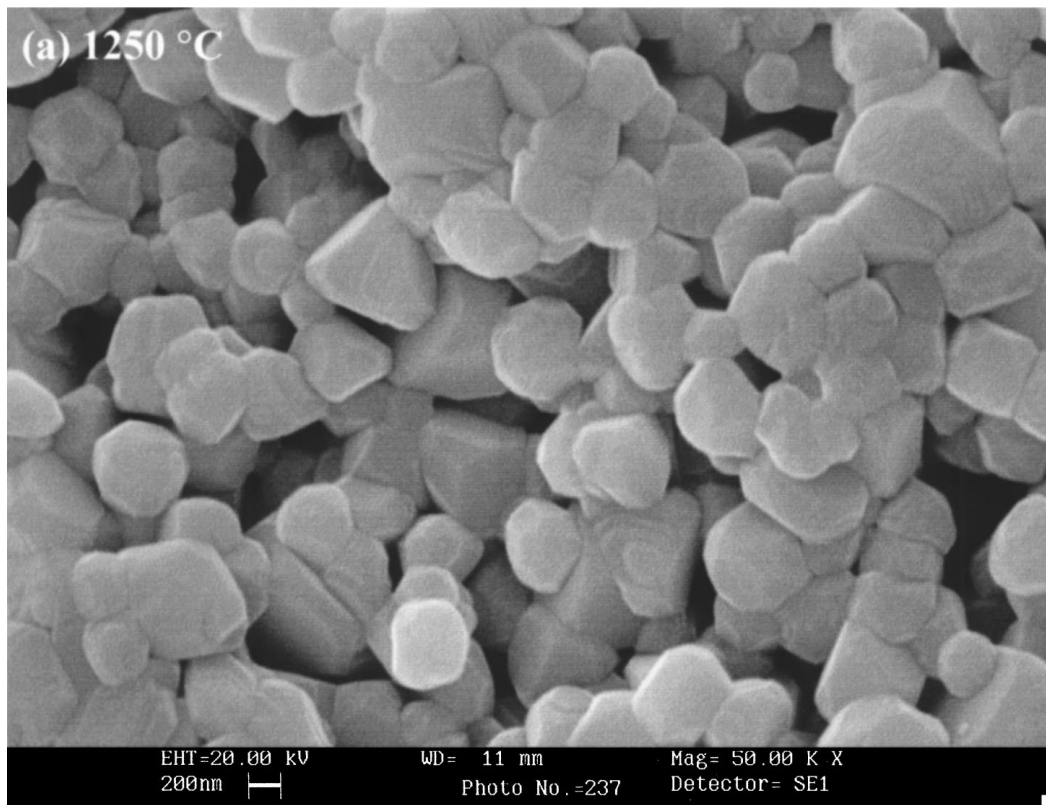


Figure 3 SEM micrographs of BaTiO<sub>3</sub> (coprecipitation) ceramic sintered at (a) 1250°C, (b) 1280°C and (c) 1300°C and (d) 1320°C. (Continued.)

the XRD patterns of BaTiO<sub>3</sub> (sol-gel) and BaTiO<sub>3</sub> (oxide) are very similar to that shown in Fig. 5 showing that perovskite structure has been formed. The tetragonality ( $c/a$  ratio) are obtained from the XRD patterns and shown in Table I. The tetragonality of both BaTiO<sub>3</sub> (sol-gel) and BaTiO<sub>3</sub> (coprecipitation) increases as  $T_s$

increases and saturates at 1.010, a value closed to the reported value [10–12]. The densities of the ceramic samples are also given in Table I. In subsequent work, properties of BaTiO<sub>3</sub> (sol-gel), BaTiO<sub>3</sub> (coprecipitation) and BaTiO<sub>3</sub> (oxide) sintered at 1320°C are reported and compared.

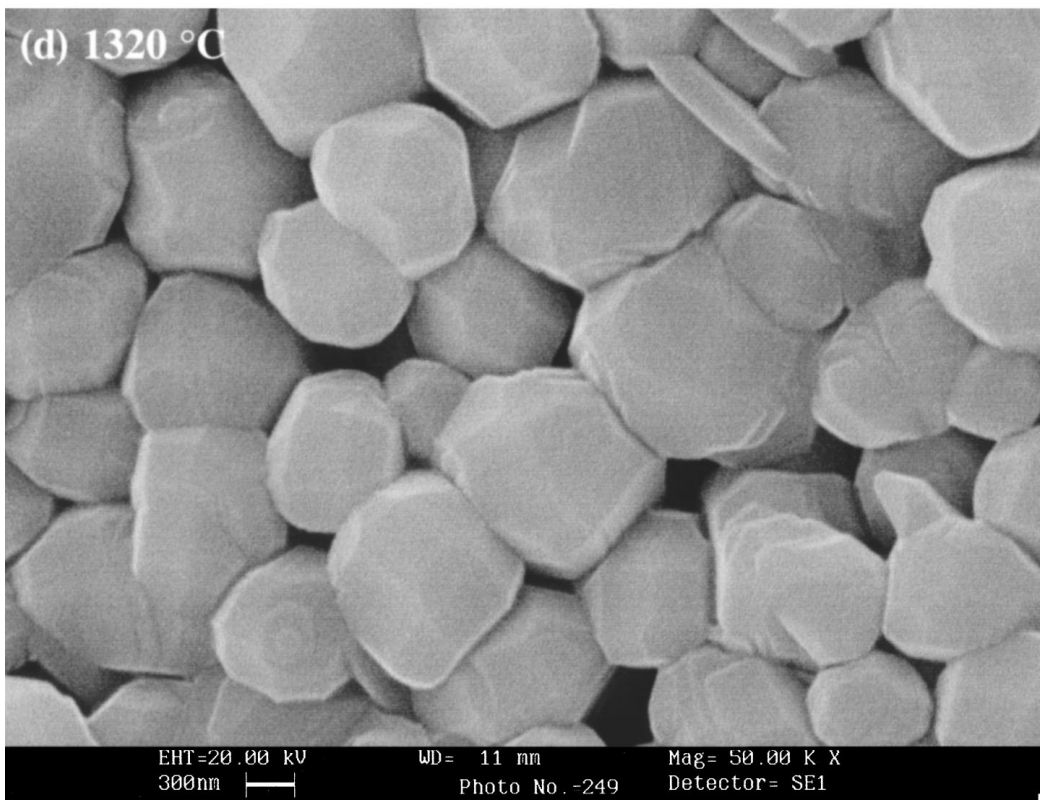
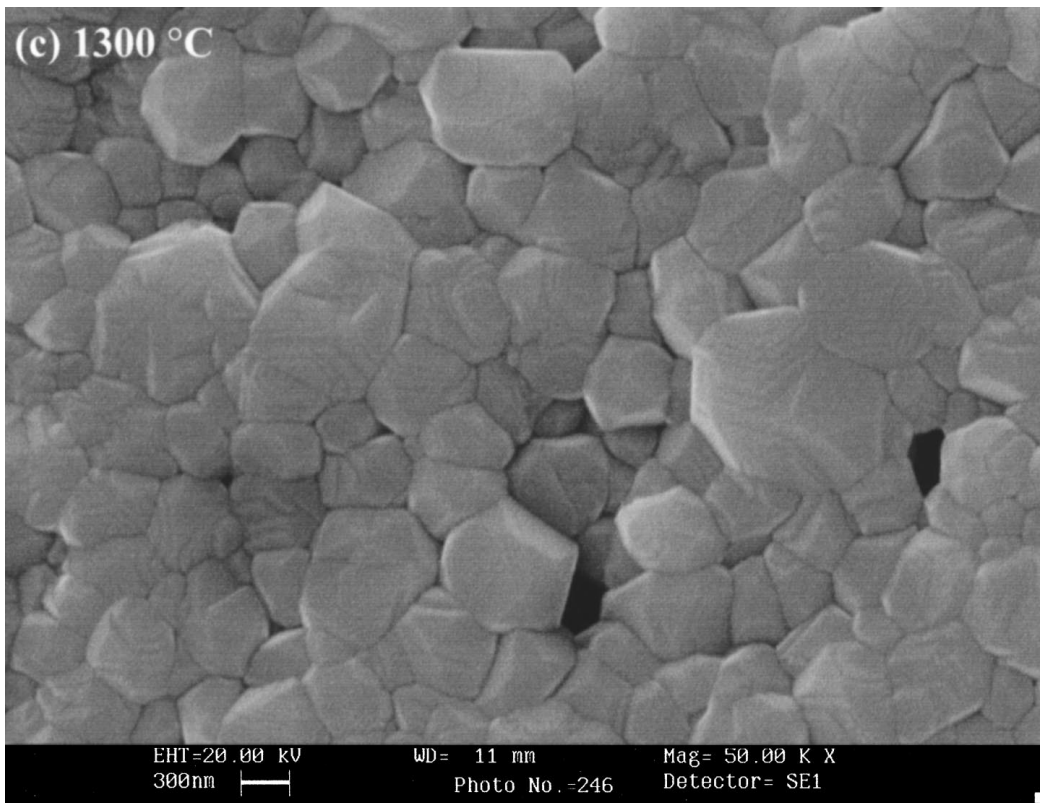


Figure 3 (Continued).

### 3.2. Dielectric, pyroelectric and piezoelectric properties

All BaTiO<sub>3</sub> samples were poled at 90 °C in an oil bath by applying various d.c. fields. After poling, the samples were annealed in a short-circuited condition at 50 °C for 3 h before measurements. This process can eliminate the contribution of thermally stimulated current in subsequent measurements. The dielectric per-

mittivity  $\epsilon$  and loss  $\tan \delta$  of the samples were measured before poling. The samples were poled using different poling fields and as the poling field increases,  $\epsilon'$  decrease (Fig. 6) which may be due to the domain orientation after poling. As the grain size of the ceramic decreases (Table I),  $\epsilon'$  increases (Table II). which exhibit a grain size effect similar to that reported in the literature [10, 11].

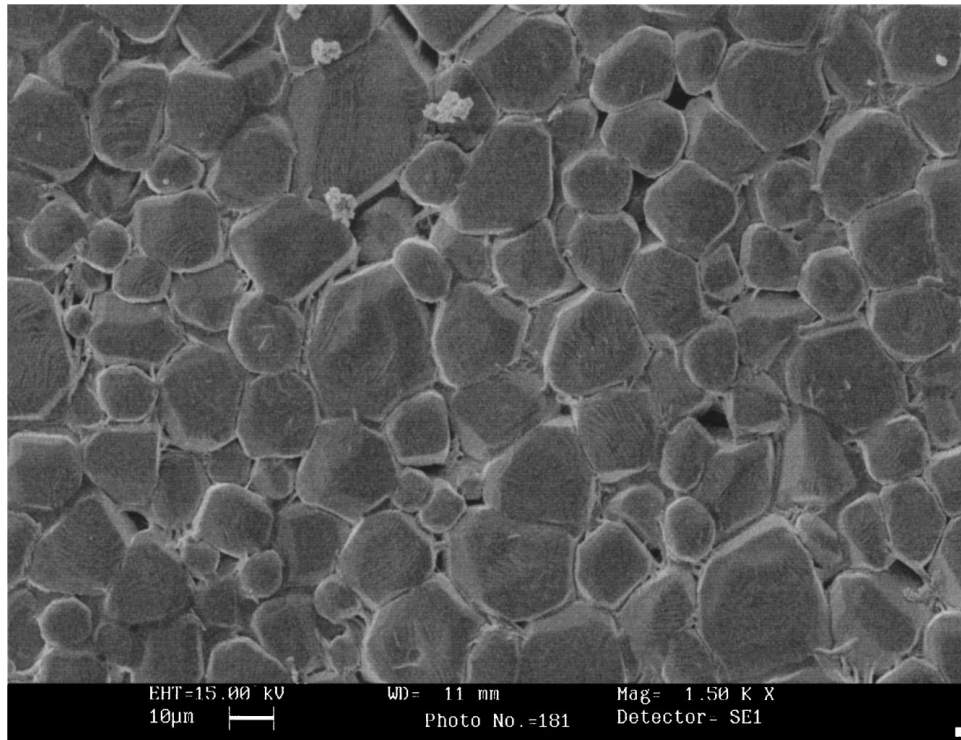


Figure 4 SEM micrograph of BaTiO<sub>3</sub> (oxide) ceramic sintered at 1320°C.

TABLE II Properties of BaTiO<sub>3</sub> ceramics after being poled with  $E = 5$  kV/mm at 90°C

Synthesis method	Sol-gel	Coprecipitation	Mixed oxide
Sintering temperature (°C)	1320	1320	1320
Pyroelectric coefficient $p$ ( $\mu\text{C}/\text{m}^2\text{K}$ )	121	157	235
Piezoelectric coefficient $d_{33}$ (pC/N)	117	143	175
Thickness electromechanical coupling coefficient $k_t$	0.16	0.22	0.25
Dielectric permittivity $\epsilon'$ (1 kHz, before poling)	1825	1941	1160
Dielectric loss $\tan \delta$ (1 kHz, before poling)	0.04	0.04	0.02

The pyroelectric coefficient  $p$  and piezoelectric coefficient  $d_{33}$  of the BaTiO<sub>3</sub> ceramics were measured at 25°C as functions of the poling field in Figs 7 and 8, respectively. Both  $p$  and  $d_{33}$  increase as the poling field increases and tend to saturate when the poling field is above 3 kV/mm. The maximum attainable  $p$  and  $d_{33}$  are given in Table II. BaTiO<sub>3</sub> (oxide) has the maximum  $p$  and  $d_{33}$  which are closed to the reported values of  $p = 200 \mu\text{C}/\text{m}^2\text{K}$  and  $d_{33} = 190 \text{ pC}/\text{N}$  [7, 12]. Fig. 9 shows the relationship between the thickness electromechanical coupling coefficient  $k_t$  and the poling field. Similar to  $p$  and  $d_{33}$ ,  $k_t$  increases as the poling field increases and tends to saturate when the samples is poled above  $E = 3$  kV/mm. The maximum attainable  $k_t$  are also given in Table II. However, the measured  $k_t$  of the BaTiO<sub>3</sub> ceramics is smaller than the reported value ( $k_t \sim 0.35$  [1, 7]). BaTiO<sub>3</sub> (coprecipitation) has  $k_t$  close to BaTiO<sub>3</sub> (oxide) and is higher than that of BaTiO<sub>3</sub> (sol-gel).

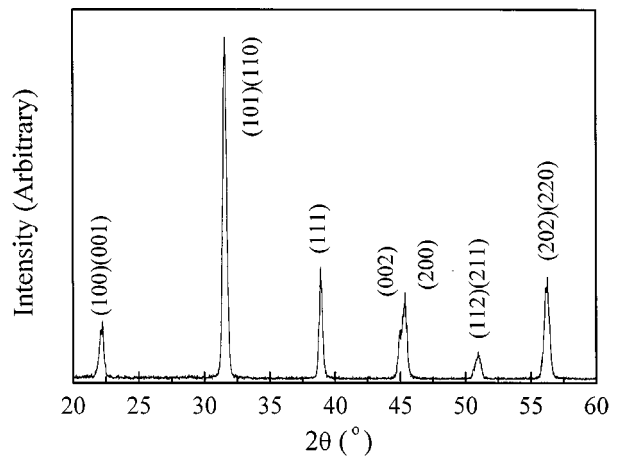


Figure 5 XRD patterns of BaTiO<sub>3</sub> (coprecipitation) ceramic sintered at 1320°C.

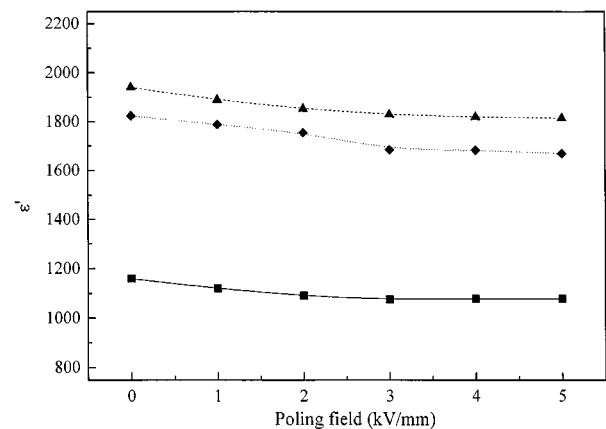


Figure 6 Room temperature dielectric permittivity  $\epsilon'$  of BaTiO<sub>3</sub> sintered at 1320°C versus poling field.



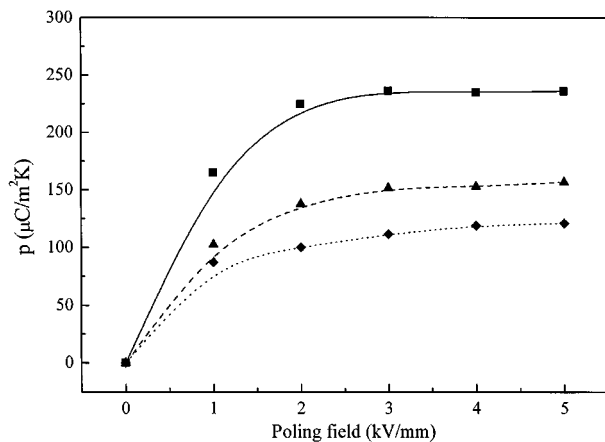


Figure 7 Pyroelectric coefficient  $p$  of  $\text{BaTiO}_3$  sintered at  $1320^\circ\text{C}$  versus poling field.

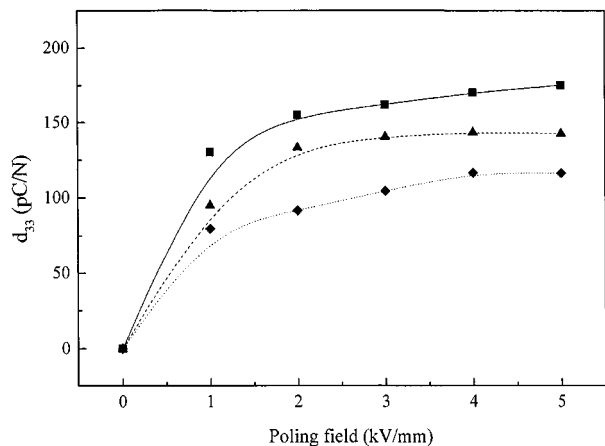


Figure 8 Piezoelectric coefficient  $d_{33}$  of  $\text{BaTiO}_3$  sintered at  $1320^\circ\text{C}$  versus poling field.

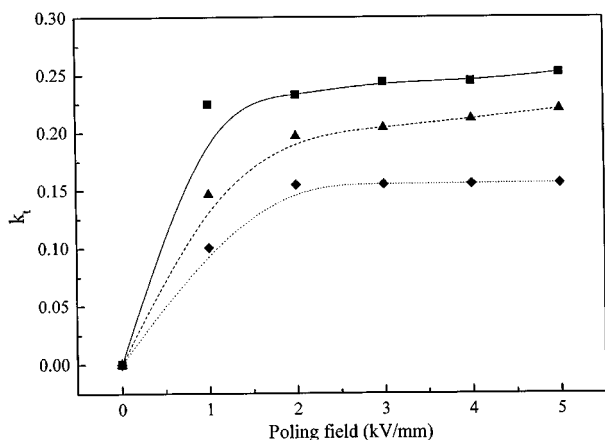


Figure 9 Thickness electromechanical coupling coefficient  $k_t$  of  $\text{BaTiO}_3$  sintered at  $1320^\circ\text{C}$  versus poling field.

#### 4. Conclusion and discussion

$\text{BaTiO}_3$  ceramics have been fabricated using powder prepared by the sol-gel, coprecipitation and conventional mixed oxide methods. Characteristics of  $\text{BaTiO}_3$  ceramics prepared by these methods have been investigated. As expected,  $\text{BaTiO}_3$  (oxide) has the largest grain

size ( $20\ \mu\text{m}$ ) and the best properties. The average grain size of  $\text{BaTiO}_3$  (sol-gel) sintered at  $1320^\circ\text{C}$  is about  $10\ \mu\text{m}$  which is too large for thick film applications and it also breaks down easily upon poling. After poling with d.c. field higher than  $3\ \text{kV/mm}$ , the pyroelectric coefficient  $p$ , piezoelectric coefficient  $d_{33}$  and the thickness electromechanical coupling coefficient  $k_t$  of  $\text{BaTiO}_3$  (sol-gel) tend to saturate. Although not shown in detail, we found that if  $\text{BaTiO}_3$  (sol-gel) was sintered at  $1200^\circ\text{C}$ , except for having a smaller grain size ( $5\ \mu\text{m}$ ) and a higher dielectric permittivity, the measured values of  $p$ ,  $d_{33}$  and  $k_t$  are almost identical to  $\text{BaTiO}_3$  (sol-gel) sintered at  $1320^\circ\text{C}$  showing that there is an upper limit of materials properties which cannot be improved by the grain growth. For  $\text{BaTiO}_3$  (coprecipitation), as the sintering temperature increases, the grain growth is very slow. The  $\text{BaTiO}_3$  (coprecipitation) sample sintered at  $1320^\circ\text{C}$  has an average grain size of  $1.13\ \mu\text{m}$  which is much smaller than  $\text{BaTiO}_3$  (sol-gel) and  $\text{BaTiO}_3$  (oxide).

In conclusion, it is found that  $\text{BaTiO}_3$  ceramic sintered by using nanosized powder derived from the coprecipitation method has good potential to be used in thick film device applications.  $\text{BaTiO}_3$  (coprecipitation) has a smaller grain size, its  $p$  and  $d_{33}$  are higher than those of  $\text{BaTiO}_3$  (sol-gel) and its  $k_t$  is comparable to that of  $\text{BaTiO}_3$  (oxide).

#### Acknowledgement

Financial support from the Industrial Support Fund (ISF) of the Hong Kong Special Administrative Region (Project No. AF/147/98) and the PolyU Centre for Smart Materials are gratefully acknowledged.

#### References

1. G. H. HAERTLING, *J. Amer. Ceram. Soc.* **82** (1999) 797.
2. R. E. JONES, P. D. MAINAR, J. O. OLOWOLAFE, J. O. CAMPBELL and C. J. MOGAB, *Appl. Phys. Lett.* **60** (1992) 1022.
3. M. H. FREY and D. A. PAYNE, *ibid.* **63** (1963) 2753.
4. H. HU and S. B. KRUPANIDHI, *J. Appl. Phys.* **74** (1993) 3373.
5. M. A. AEGERTER, Y. CHARBOUILLOT, N. MOHALLEM and L. H. DE GODOY, "Ultrastructure Processing of Advanced Materials" (A Wiley-Interscience Publication, New York, 1992) p. 613.
6. T. T. FANG, H. B. LIN and J. B. HWANG, *J. Amer. Ceram. Soc.* **73** (1990) 3363.
7. B. JAFFE, W. R. COOK and H. JAFFE, "Piezoelectric Ceramics" (Academic Press Inc. Ltd., London, 1971) p. 53.
8. Y. XU, "Ferroelectric Materials and Their Applications" (Elsevier Science Publishers B. V., Amsterdam, 1991) p. 101.
9. IEEE Standard on Piezoelectricity, IEEE Std. 176-1987 (The Institute of Electrical and Electronics Engineers, New York, 1987).
10. M. P. MCNEAL, S. J. JANG and R. E. NEWNHAM, *J. Appl. Phys.* **83** (1998) 3288.
11. G. ARLT, D. HENNING and G. DE WITH, *ibid.* **58** (1985) 1619.
12. B. JAFFE, W. R. COOK and H. JAFFE, "Piezoelectric Ceramics" (Academic Press Inc. Ltd., London, 1971) p. 23.

Received 29 November 1999  
and accepted 11 May 2000